

## Study of MoO<sub>3</sub> doping effect on linear and Non-linear optical properties of Cellulose acetate (CA) films

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### A B S T R A C T

The goal of this study was to investigate the optical properties of the prepared CA-MoO<sub>3</sub> nanocomposites films. CA has been chosen as a host matrix and then doped with MoO<sub>3</sub> with different concentrations (2.5, 5, 10 and 15wt. %) of MoO<sub>3</sub> by using casting procedure. CA-MoO<sub>3</sub> films were characterized by UV-Visible spectrophotometer within the wavelength range of 200 nm-800 nm. The Linear optical parameters (transmission (T), extinction coefficient (K), reflectance (R), refractive index (n), real ( $\epsilon_1$ ) and imaginary ( $\epsilon_2$ ) part of the dielectric constant and optical conductivity ( $\sigma_{opt}$ )) of CA-MoO<sub>3</sub> films were obtained from absorption spectra. The volume (VELF) and surface (SELF) energy loss function of CA-MoO<sub>3</sub> films were obtained by complex dielectric constant. The Dispersion parameters (dispersion energy ( $E_d$ ), oscillation energy ( $E_0$ ), static dielectric constant ( $\epsilon_0$ ) and static refractive index ( $n_0$ )) were calculated using theoretical Wemple-DiDomenico model. The nonlinear optical susceptibility  $x^{(1)}$ ,  $x^{(3)}$  and nonlinear refractive index  $n_2$  were evaluated from the linear optical parameters using semiempirical relation. The results show that the Linear and nonlinear optical parameters change with the increase of molybdenum trioxide concentration, where, the transmittance, oscillator energy, the volume (VELF) and surface (SELF) energy loss decreases with the increase of MoO<sub>3</sub> concentration.

**Keywords:** CA; SELF; static dielectric constant; refractive index; oscillation energy.

### 1. Introduction

Polymer composites are important because of their valuable role in various aspects of our daily lives. Their attractive characteristics, including their low-cost, abundance, and flexibility. Most polymers are considered non-toxic and could easily play the role of host matrices for different sorts of fillers [1–8]. Improving the polymers' optical, electrical and mechanical properties can be achieved via implanted limited concentrations of fillers to match desired applications. These applications include optoelectronic

devices, sensors, solar cells, capacitors, reduction of environmental pollution and ceramics [9]. Natural polymers like cellulose show a great potential for future use as an alternative for the non-biodegradable polymers their low energy consumption. Cellulose acetate (CA) in particular of all other cellulose derivatives because of its excellent optical clarity, biodegradability, and high toughness and other physical properties. CA is also largely used as a film base in photography and as a filler material in cigarettes. Due to its low cost can be used in textile, membrane technology, drug release, UV protector, Opto-electronic field and a wide range of applications in different fields. This wide range application of CA is limited by its high

moisture sensitivity and brittle nature. Studies show that to improve CA properties, nanomaterials such as graphene oxide, montmorillonite, carbon nanotubes, silver, and gold are embedded in CA matrix, can reinforce and enhance its properties effectively, such as UV shields, flame retardants, photo activity and gas permeability [10].

Transition metal oxides such as MoO<sub>3</sub> are semiconductor materials that have a wide technological application, in display devices, optical smart windows, Electrochromic devices, and gas sensors. MoO<sub>3</sub>, exhibit great structural, chemical, and optical properties. It is considered as a cryogenic material because it displays electrophotochromic and gasochromic properties by virtue of which gives the material the potential for the development of electronic display devices. MoO<sub>3</sub> film has a several application, as in sensors and lubricants, and makes it a candidate for back contact layer of cadmium Telluride solar cells [11].

In the present work, the linear optical parameters of CA-MoO<sub>3</sub> films have been evaluated from absorption spectra. As per author's information, first time the dispersion of refractive index and the non-linear optical parameters of CA-MoO<sub>3</sub> were studied using semi empirical models.

## 2. Experimental methods

### 2.1. Materials

Cellulose acetate polymer (CA) was used as starting material for preparation of films. It was purchased from Thomas Baker Lab Chemicals, India, white color and nontoxic powder has 53.5-54.5% degree of acetylation. Molybdenum trioxide (MoO<sub>3</sub>) was obtained from Riedel-de Haen, (Germany), as a light gray powder, with a molecular weight of 143.94 g/Mol, 99.5% purity and was used without further purification. Dichloromethane compound (M.W=84.93) with a purity of 99%, and N, Ndimethyl formamide (M.W=73.9) with a purity of 99% were obtained from the Park Scientific Limited, Northampton U.K. It was used without any further purification and employed to dissolve CA and MoO<sub>3</sub>.

### 2.2. Preparation of composite films

Cellulose acetate (CA) with different proportions of molybdenum trioxide (MoO<sub>3</sub>) (2.5, 5, 10 and 15wt. %) films were prepared by solution casting technique. In this process, one gram of CA was dissolved in the volume ratio of (25:1) dichloromethane: N, N-dimethyl formamide. The mixture is placed on a magnetic stirrer to complete the solubility with continued stirring at room temperature. An exact weight of MoO<sub>3</sub> based on the weight of CA was added to the solution of CA. After MoO<sub>3</sub> completely dissolved in CA solution, the mixture was poured into a flat bottom glass container prior to

prepare the film. The glass container was covered and placed on a flat bench to evaporate both dichloromethane and N, N-dimethyl formamide solvent.

### 2.3. Optical characterization

A UV-visible spectrophotometer (Lambda 25, Perkin Elmer) was utilized to record the absorbance (A) and transmittance (T) spectra of the prepared samples over the scanned wavelength range 200–800 nm at room temperature. Based on the measured A and T data, the extinction coefficient (K), refractive index (n), reflectance (R) and the optical conductivity (σ<sub>opt</sub>) values of the prepared composites were determined using the following equations [12–19]:

$$k = \frac{\alpha\lambda}{4\pi} \quad (1)$$

$$n = \frac{1 + \sqrt{R}}{1 - \sqrt{R}} + \sqrt{\frac{4R}{(1 - R)^2} - K^2} \quad (2)$$

$$R = 1 - \sqrt{T - e^A} \quad (3)$$

$$\sigma_{op} = \frac{\alpha nc}{4\pi} \quad (4)$$

Where (α) is the absorption coefficient, (c) is the speed of light in a vacuum and (λ) is the wavelength of the incident light.

The fundamental electron excitation spectra of CA-MoO<sub>3</sub> films are described with complex dielectric constant. The real part of dielectric constant is related to the property of slowing down the speed of light in the materials. Whereas, the imaginary part of dielectric constant is related with the absorbing energy from electric field due to dipole motion. The value of real and imaginary part of dielectric constant was obtained from the following formula; [20]:

$$\epsilon = \epsilon_1 + \epsilon_2 \quad (5)$$

Where

$$\epsilon_1 = n^2 - k^2 \quad (6)$$

$$\epsilon_2 = 2nk \quad (7)$$

The energy dispersion parameter of refractive index can be evaluated from the supposed Wemple and DiDomenico (WDD) single-oscillator model according to the following equation [21]:

$$(n^2 - 1)^{-1} = \frac{E_0}{E_d} + \frac{1}{E_0 E_d} (hv)^2 \quad (8)$$

Where  $E_0$  is the average value of the single oscillator energy of the electronic transitions, which is usually considered as an average energy gap, and  $E_d$  is the dispersion energy that measures the average strength of inter-band optical transitions. The static refractive index  $n_0$  at a zero photon energy is determined from Eq. (9) in Ref. [22]:

$$n_0^2 = 1 + \frac{E_d}{E_0} \quad (9)$$

The energy loss of the incident electromagnetic waves by electrons in a material or surface coating is connected with the volume energy loss function (VELF) and the surface energy loss function (SELF), respectively. Both the surface and volume energies are obtained from complex dielectric constant [23]:

$$SELF = \frac{\epsilon_2}{(\epsilon_1 + 1)^2 + \epsilon_2^2} \quad (10)$$

$$VELF = \frac{\epsilon_2}{\epsilon_1^2 + \epsilon_2^2} \quad (11)$$

Nonlinear optical parameters of materials are very useful for the fabrication of frequency conversion and optical switching devices, which in turn play an essential role in Photonics. The third order optical non-linear susceptibility gives the information about the strength of chemical bonds between the molecules of CA-MoO<sub>3</sub> films [24]. The linear optical susceptibility  $\chi^{(1)}$  of the medium can be computed from the relation [25]:

$$\chi^{(1)} = \frac{E_d/E_0}{4\pi} \quad (12)$$

The nonlinear third order of the optical susceptibility  $\chi^{(3)}$  could be estimated from linear optical susceptibility by applying  $\chi^{(1)}$  Miller's rule like the following formula [25]:

$$\chi^{(3)} = k (\chi^{(1)})^4 \quad (13)$$

Where,  $k$  is constant ( $1.7 \times 10^{-10}$  esu).

### 3. Results and discussion

From the transmittance spectra Figure 1, it can be observed that the transmittance decreases as the doping wt% is increased.

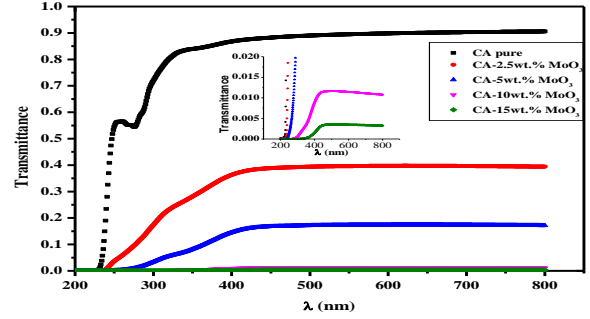


Fig.1. The transmittance spectra for CA-MoO<sub>3</sub> films as a function of incident wavelength.

Figure 2 presents the variations of extinction coefficient ( $k$ ) with the wavelength for pure CA and CA-MoO<sub>3</sub> composite films. It can be seen that the value of  $k$  increases with MoO<sub>3</sub> weight percentage increase. The increase of the extinction coefficient for the electrolyte films with MoO<sub>3</sub> concentration is due to the increase in the absorption coefficient [26]. This result indicates that the doping of MoO<sub>3</sub> tends to change the optical properties of the host polymer.

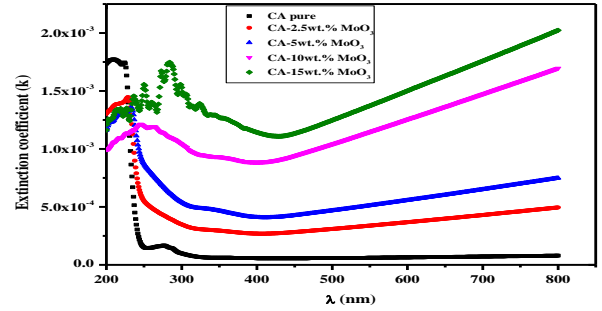


Fig.2. The variation of the extinction coefficient ( $k$ ) with wavelength of the CA and CA-MoO<sub>3</sub> composite films.

The reflectance increases with an increase in the concentration of the doped MoO<sub>3</sub> nanoparticles; see Figure 3.

Figure 4 displays the increase in refractive index of the doped MoO<sub>3</sub> films with increasing the content of MoO<sub>3</sub> nanoparticles, with respect to pure CA. It is observed also that the refractive index reduced at higher energy (shorter wavelength), which suggests the behavior of the normal dispersion of the material in the visible range [27,28].

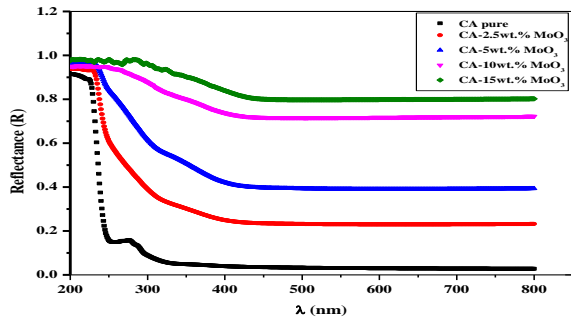


Fig.3. The reflectance of CA and CA-MoO<sub>3</sub> films as a function of wavelength.

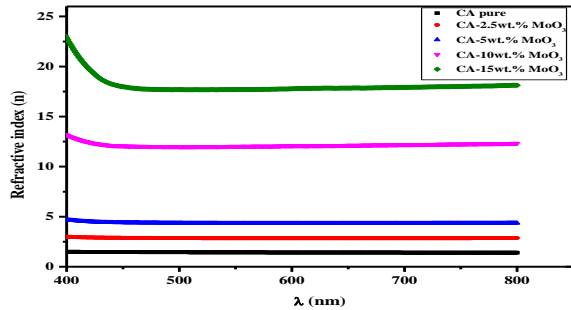


Fig.4. The refraction index of prepared films as a function of wavelength.

Figure 5 shows the variation of optical conductivity of the CA - MoO<sub>3</sub> with the wavelength of the incident photon. It is clear that the optical conductivity of the films increases with the increase in MoO<sub>3</sub> weight percentage. This increase can be attributed to the creation of new energy levels in the band gap which facilitate the crossing of electrons from the valence band to these local levels, then to the conduction band and, consequently, the band gap decreases and the conductivity increases [29].

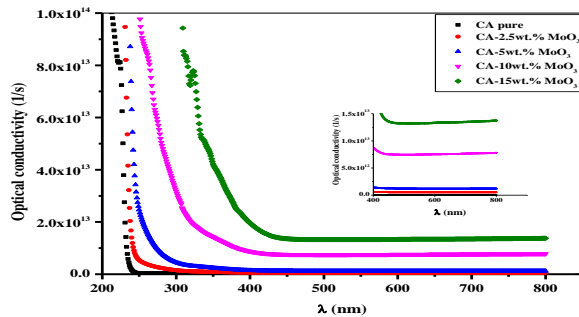


Fig.5. The optical conductivity of the prepared films as a function of incident wavelength.

The values of  $E_0$  and  $E_d$  were calculated by plotting  $(n^2-1)^{-1}$  versus  $(h\nu)^2$  as shown in Figure 6. The plots give straight lines which intercept y-axis in a value equal to  $(E_0/E_d)$ , and the slope equal to  $(-1/E_0E_d)$ . The evaluated values of  $E_0$  and  $E_d$  are recorded in Table 1. It is obvious that the single-oscillator energy,  $E_0$  values decrease with increasing MoO<sub>3</sub> content and  $E_d$  values were enhanced by increasing the MoO<sub>3</sub> concentration. The increase in

the dispersion energy ( $E_d$ ) indicates the increase in bond strength, which leads to increase in the degree of disorder [21].

The computed values of the static refractive index ( $n_0$ ) and the static dielectric constant ( $\epsilon_0 = n_0^2$ ) are listed in Table.1 which were enhanced with the concentration of MoO<sub>3</sub> nanoparticles.

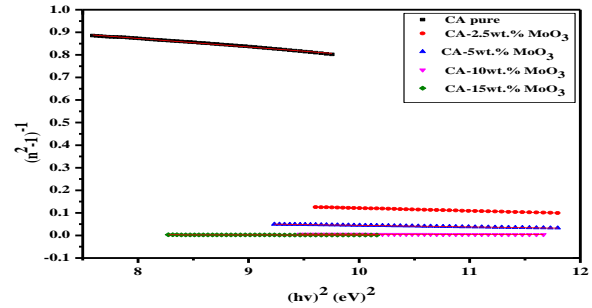


Fig.6. The relationship between  $(n^2-1)^{-1}$  and  $(h\nu)^2$  for CA and CA - MoO<sub>3</sub> films.

Table1. Single-oscillator energy,  $E_0$ , dispersion energy,  $E_d$ , static refractive index ( $n_0$ ) and the static dielectric constant,  $\epsilon_0$ , values according to Wemple and DiDomenico model.

Sample (wt%)	$E_0$ (eV)	$E_d$ (eV)	$n_0$	$\epsilon_0$
0	5.58	4.75	1.36	1.85
2.5	4.47	18.47	2.27	5.15
5	4.07	36.10	3.14	9.86
10	3.90	248.4	8.04	64.64
15	3.53	403.43	10.74	115.34

The dependence of  $\epsilon_1$  and  $\epsilon_2$  on photon wavelength is shown in Figures 7 and 8. It is observed that the real part values of dielectric constant  $\epsilon_1$  depend more than the imaginary part of dielectric constant  $\epsilon_2$  which is due to  $\epsilon_1$  on the refractive index ( $n$ ), on the other hand  $\epsilon_2$  depends on the extinction coefficient ( $k$ ) [20]. The values of both parameters increase with the increase of the weight percentage of MoO<sub>3</sub> nanoparticles in the CA matrix.

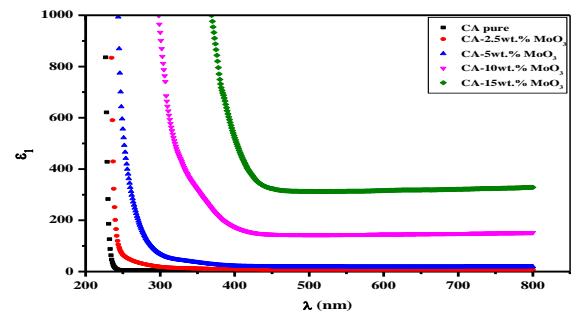


Fig.7. The real dielectric constant for CA and CA-MoO<sub>3</sub> composite films as a function of incident wavelength.

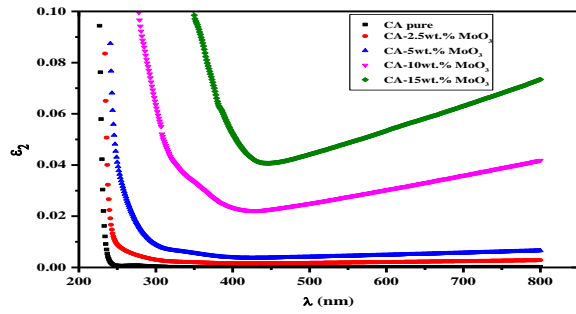


Fig.8. The imaginary dielectric constant of CA and CA-MoO<sub>3</sub> composites films as a function of incident wavelength.

The variations of VELF and SELF with the incident spectrum energy in the energy range of (1.5–6.5 eV) are shown in Figures 9 and 10, respectively. The VELF values are always greater than SELF values for all the samples. The VELF and SELF decreased as the MoO<sub>3</sub> nanoparticle concentration increased.

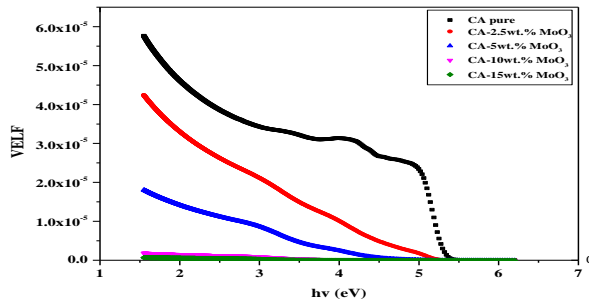


Fig.9. Variation of a VELF with the photon energy for the CA-MoO<sub>3</sub> nanocomposites.

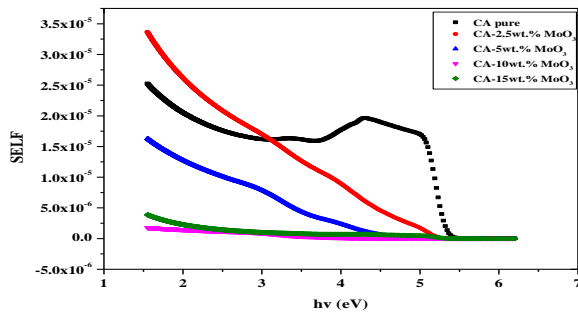


Fig.10. Variation of a SELF with the photon energy for the CA-MoO<sub>3</sub> nanocomposites.

The evaluated values of the linear optical susceptibility are listed in Table 2. It was found that the values of the linear optical susceptibility, nonlinear third-order optical susceptibility, and nonlinear refractive index increase as the content of MoO<sub>3</sub> nanoparticles increases. The high values of the nonlinear third-order susceptibility and nonlinear refractive index show the suitability of the samples to be used in different nonlinear optical and photonic devices and applications [25].

Table 2. Values of  $x^{(1)}$ ,  $x^{(3)}$ , and  $n_2$  of the prepared CA-MoO<sub>3</sub> nanocomposites

Sample (wt%)	$x^{(1)}$	$x^{(3)}$ (e.s.u)	$n_2$ (e.s.u)
0	0.0678	$3.59 \times 10^{-15}$	$9.95 \times 10^{-14}$
2.5	0.329	$1.99 \times 10^{-12}$	$3.30 \times 10^{-11}$
5	0.706	$4.22 \times 10^{-11}$	$5.06 \times 10^{-10}$
10	5.071	$6.61 \times 10^{-8}$	$3.1 \times 10^{-7}$
15	9.099	$1.17 \times 10^{-6}$	$4.1 \times 10^{-6}$

#### 4. Conclusion

Linear and nonlinear optical studies were performed for pure CA and CA doped by MoO<sub>3</sub> with different concentrations utilizing the casting techniques. The transmittance (T) and the average values of single oscillator energy for electronic transitions ( $E_0$ ) are decreasing with increasing MoO<sub>3</sub>. The extinction coefficient (K), refractive index (n), reflectance (R) and the optical conductivity ( $\sigma_{opt}$ ), refractive index (n), and dielectric constant ( $\epsilon$ ) increase with increasing MoO<sub>3</sub>. Third-order nonlinear optical susceptibility  $x^{(3)}$  and the nonlinear refractive index  $n_2$  were observed to increase with increasing the amount of MoO<sub>3</sub> nanoparticles.

These results indicate that the film we had prepared can be used in many applications, including anti-reflective coating, photonic devices and as a UV sensor.

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